

## A molten battery consisting of Li metal anode, AlCl<sub>3</sub>-LiCl cathode and solid electrolyte



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### ABSTRACT

High performance rechargeable batteries are urgently needed to address the demands of grid-scale stationary energy storage. High temperature battery systems, such as Na-S battery, Na-NiCl<sub>2</sub> battery (ZEBRA battery) and liquid metal electrode (LME) battery, exhibit advantages like high power density and high cyclic stability, but also suffer from the high operating temperature. We recently invented new concept of molten lithium metal batteries, consisting of liquid lithium anodes, alloy (Sn, Bi, Pb) liquid cathodes and lithium ion conductor as solid electrolytes. Here we demonstrate a molten metal chloride battery that operates at a relatively low temperature of 210 °C. The battery has been designed to include molten (AlCl<sub>3</sub>-LiCl) cathode, solid electrolyte (garnet-type Li<sub>6.4</sub>La<sub>3</sub>Ta<sub>0.6</sub>Zr<sub>1.4</sub>O<sub>12</sub> (LLZTO) ceramic tube) and molten lithium anode. The assembled AlCl<sub>3</sub>-LiCl||LLZTO||Li full cell has an average discharge voltage of 1.55 V and energy efficiency of 83% and has been successfully cycled for 100 cycles (800 h) without capacity fade. The theoretical specific energy of cell is 350 Wh/kg and the estimated cost is \$11.6/kWh based on the weight of electrode materials. Considering the high performance, high safety, low operating temperature and low cost of raw materials, our new type of molten-electrode battery system opens up new opportunities for stationary energy storage.

Large-scale electrochemical energy storage system is critical for the renewable energy and smart grid technologies [1–3]. In particular, rechargeable batteries with low cost, long lifespan, good safety and high power density are required for stationary energy storage [4–6]. Many types of batteries technologies are being developed, examples including traditional lead-acid batteries [7], Li-ion batteries [8–10], Al-ion batteries [11,12], and flow batteries [13,14]. However, such room-temperature batteries suffer from high cost or limited lifespan, which seriously restricts their applications in stationary energy storage [15,16]. Recently, high-temperature batteries, including Na-S battery [17] and liquid metal electrode (LME) battery [18], have become an area of focus since they potentially meet the performance requirements for the smart electric grid. However, the exceedingly high operation temperature (>300 °C), which results in safety hazards and high maintenance cost, still remains as serious problem. The ZEBRA battery (Na-NiCl<sub>2</sub>) can be operated at a relatively low temperature (250 °C), however, the high

cost of Ni cathode severely limits its practical applications [19,20]. Therefore, a new molten rechargeable battery for large-scale energy storage, which can be operated at a relative low temperature with an acceptable cost, is still challenging.

Recently, our team developed a new-type LME battery using LLZTO ceramic tube as solid electrolyte, which can operate at a relative low temperature of 240 °C [21]. In the first demonstration, we show a battery chemistry system consisting of molten Bi-Sn-Pb alloying cathodes and molten Li metal anodes with theoretical specific energy limited to 230 Wh/kg. Herein, we designed a new high temperature battery chemistry with much higher theoretical specific energy and lower cost, which consists of AlCl<sub>3</sub>-LiCl molten salt cathode, solid electrolyte tube (garnet-type Li<sub>6.4</sub>La<sub>3</sub>Ta<sub>0.6</sub>Zr<sub>1.4</sub>O<sub>12</sub> (LLZTO) ceramic tube) and liquid lithium anode. The AlCl<sub>3</sub>-LiCl||LLZTO||Li full cell delivered a high cyclic performance and an acceptable power/energy density with a relative low cost. Besides, the relative low operating temperature and low

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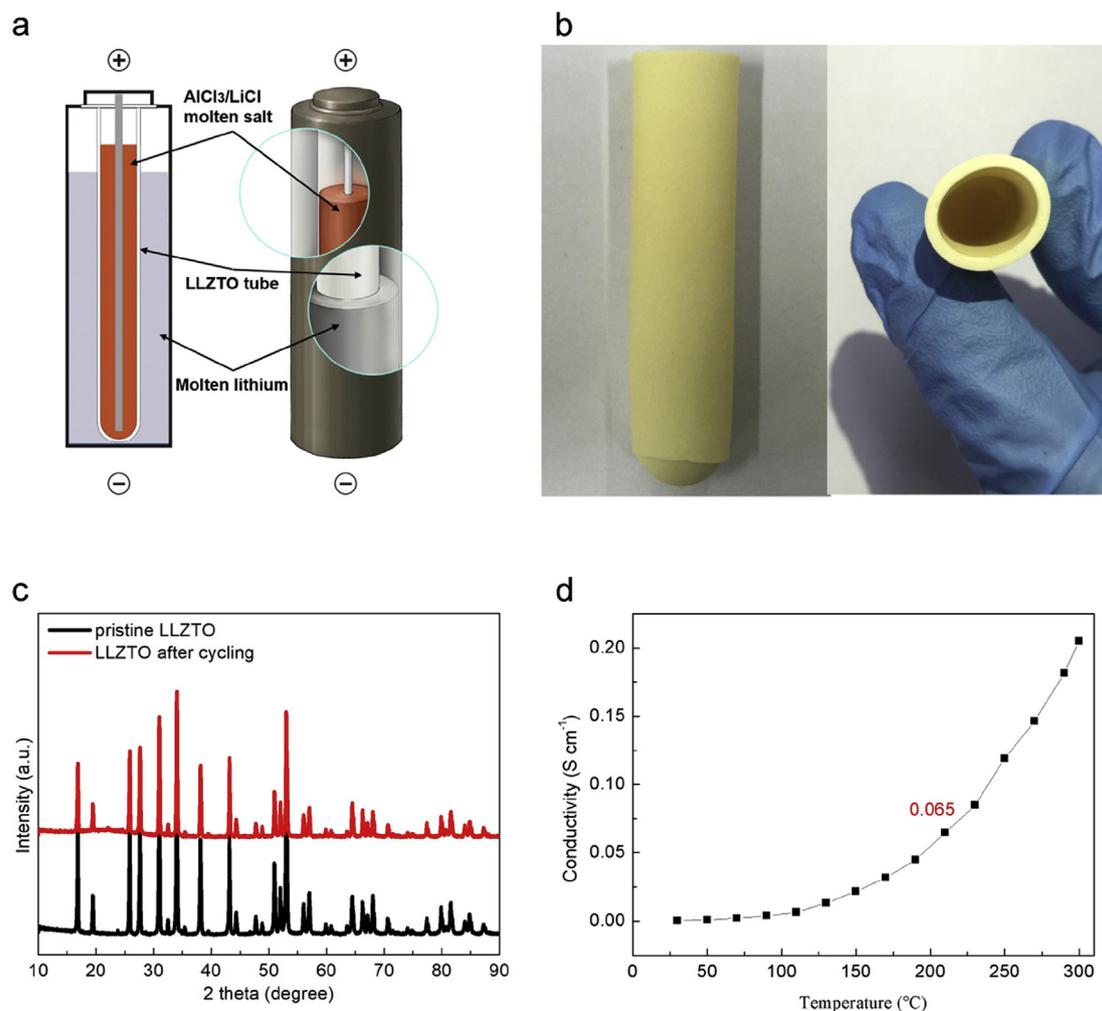


Fig. 1. (a) The construction of the Li||LLZTO||AlCl<sub>3</sub>-LiCl cell. (b) Digital photos of the LLZTO ceramic tube. (c) XRD pattern of the LLZTO ceramic tube before and after cycling. (d) Ionic conductivity of LLZTO ceramic from 30 to 300 °C.

flammability of the cathode materials ensure the high safety of the full cell. Therefore, the new-designed AlCl<sub>3</sub>-LiCl||LLZTO||Li full cell is a promising candidate for stationary energy storage.

We design the battery with AlCl<sub>3</sub>-LiCl cathode, LLZTO solid electrolyte and lithium anode. We selected the AlCl<sub>3</sub>-LiCl as cathode with the following design thinking in mind: 1) The Li-AlCl<sub>3</sub> redox couple is a promising candidate owing to its high theoretical specific capacity (520.7 mAh g<sup>-1</sup>). 2) The AlCl<sub>3</sub>-LiCl system have a low melting point of 110 °C. 3) The AlCl<sub>3</sub> materials have much lower cost compared with other metal chlorides, like NiCl<sub>2</sub>. The price of Al is only 15% of the price of Ni. 4) The corrosion of the above molten salt system is much lower than the system containing Ni<sup>2+</sup> and Fe<sup>3+</sup>, which provides more economical material choices used for cathode current collectors and cell cases [22]. The overall redox reaction of the AlCl<sub>3</sub>-LiCl||LLZTO||Li battery is as follow:

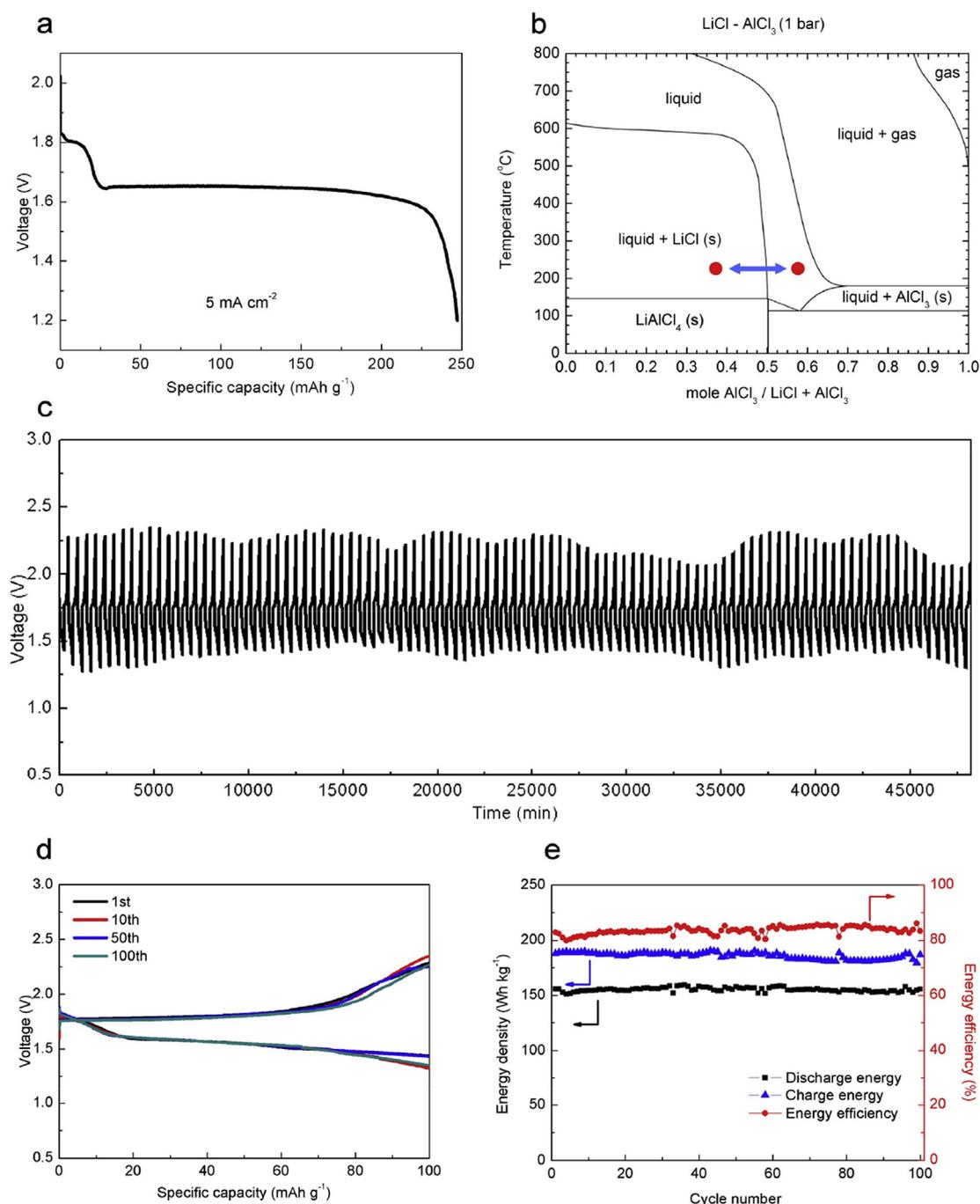


The theoretical specific energy based on the active materials is 686 Wh/kg. We emphasize that our battery system exhibits high safety owing to the relatively low operating temperature and the non-flammable cathode materials. In a Na-S battery, the failure of Al<sub>2</sub>O<sub>3</sub> electrolyte or internal shorting of battery will lead to severe safety hazard caused by the short-time high temperature and flammability of molten sulphur. As a comparison, the AlCl<sub>3</sub>-LiCl||LLZTO||Li battery system would exhibit much improved safety. Even the ceramic separators have mechanical failure, there is still no serious safety hazard. In that case, the

liquid cathode salt will get into contact with the molten Li and react to form non-flammable and non-explosive LiCl and Al, which will block further reaction.

Fig. 1a shows the configuration schematic of the battery. A stainless steel cylinder was used as negative current collector, molten Li as anode, LLZTO ceramic tube (Fig. 1b,c and Fig. S1) as electrolyte, AlCl<sub>3</sub>-LiCl (60:40 mole %) as cathode and aluminum foil as positive current collector. Both anode and cathode are liquid state when the cell operates at 210 °C, which ensures the good interface contact between the molten liquid electrodes and the solid electrolyte. At such temperature, the solid electrolyte also exhibits high ionic conductivity of 65 mS cm<sup>-1</sup> at 210 °C (Fig. 1d), which is greater than that of most liquid organic electrolytes (ex. ~10 mS cm<sup>-1</sup>, 1 M LiPF<sub>6</sub> in carbonate-based solvent at room temperature) [23]. Besides, the LLZTO tube possesses a high relative density of 99%, which can effectively prevent any leakage or crossover of the liquid electrode materials and the self-discharge [21,24–27]. The existing of LiCl in the molten salt not only lowers the melting point of the cathode materials, but also functions as a secondary electrolyte to transfer lithium ions.

The voltage profile of the first discharge was shown in Fig. 2a. There is a short plateau at ~1.8 V, which is possibly caused by the side reactions in the interfaces. The main discharge plateau was at ~1.65 V, corresponding to the cathode specific capacity of ~200 mAh g<sup>-1</sup>. After the plateau, the discharge voltage rapidly fell down to the cutoff voltage (1.2 V). There was ~50% of AlCl<sub>3</sub> contributing to the reaction. As the reaction progressed, the mole ratio of the AlCl<sub>3</sub> and LiCl continuously

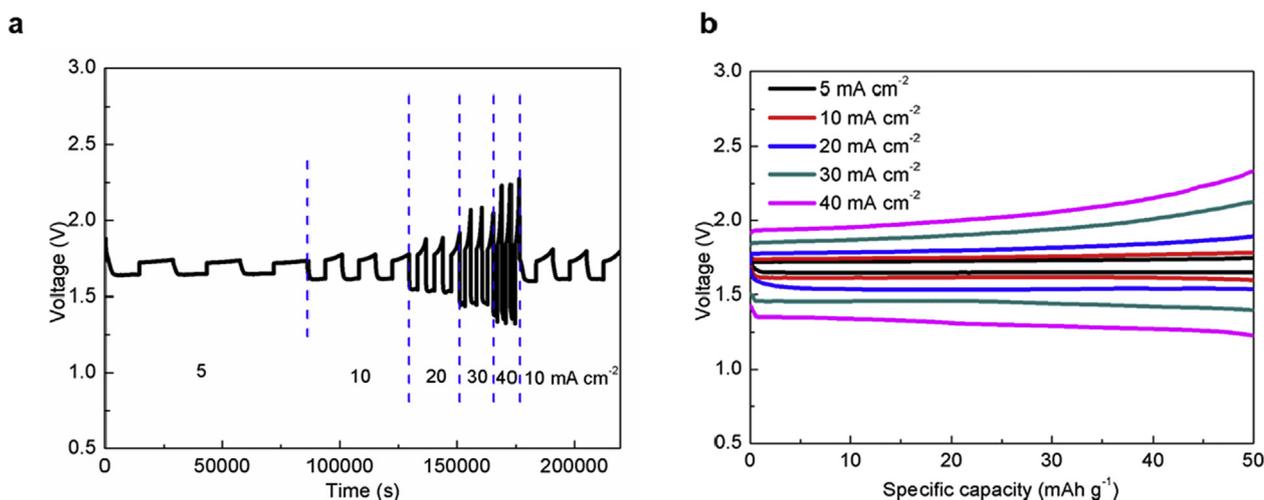


**Fig. 2.** (a) Voltage profile of the first discharge process at  $5 \text{ mA cm}^{-2}$ . (b) The phase change of cathode materials during cycling. (c) Voltage profiles from the 1st to the 100th discharge/charge cycle at  $10 \text{ mA cm}^{-2}$  with a specific capacity of  $100 \text{ mAh g}^{-1}$ . (d) Voltage profiles of the 1st, 10th, 50th and 100th discharge/charge cycles at  $10 \text{ mA cm}^{-2}$  with a specific capacity of  $100 \text{ mAh g}^{-1}$ . (e) Energy densities of cathode and energy efficiency as functions of cycle number from the 1st cycle to the 100th cycle at  $10 \text{ mA cm}^{-2}$  with a constant discharge/charge capacity of  $100 \text{ mAh g}^{-1}$ .

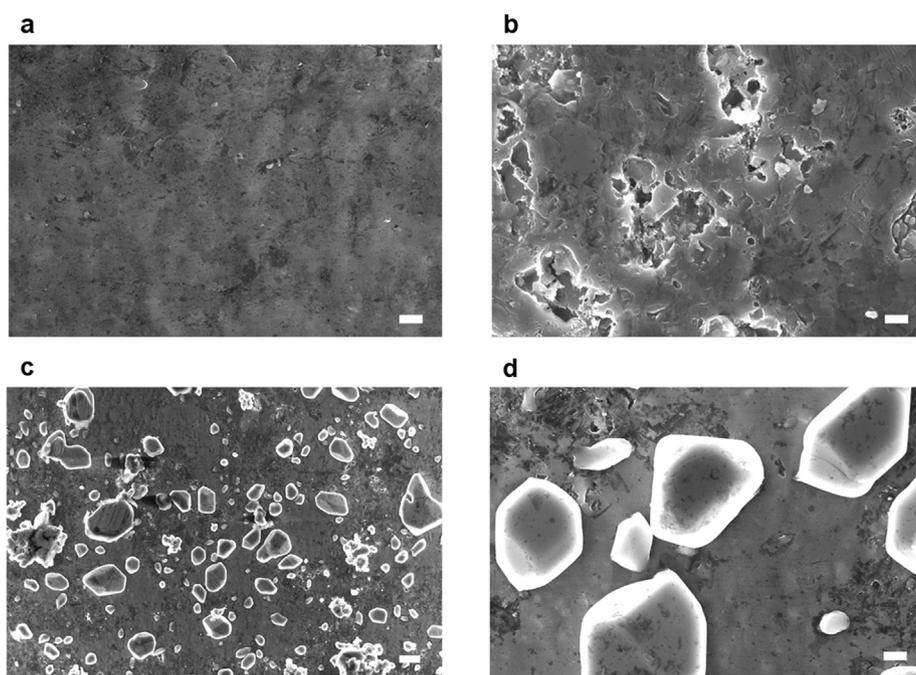
decreased, leading to the precipitation of  $\text{LiCl}$  and the reduction of the liquid phase, therefore the discharge voltage started to fall down when the specific capacity of cathode reached  $230 \text{ mAh g}^{-1}$ . Based on the weight of electrode materials, the specific energy of cell is  $350 \text{ Wh/kg}$  and the energy density per unit volume is  $584.1 \text{ Wh/L}$ . The estimated cost is  $\$11.6/\text{kWh}$ .

Long lifespan is a critical factor for large-scale energy storage applications. To test the cyclic performance, an  $\text{AlCl}_3\text{-LiCl}|\text{LLZTO}|\text{Li}$  full cell was cycled at a current density of  $10 \text{ mA cm}^{-2}$  with a constant discharge/charge capacity of  $100 \text{ mAh g}^{-1}$  for 100 cycles (800 h). The phase change of the cathode materials during cycling was shown in Fig. 2b. During the

discharge process, part of solid  $\text{LiCl}$  precipitated. Before cycling, the cell was activated to lessen the influence of interface side reactions. As shown in Fig. 2c and d, the cell exhibited a stable cycling performance. The voltage profiles of the 1st, 10th, 50th and 100th are highly consistent (Fig. 2d). The voltage upper and lower limit stabilized at around 2.2 V and 1.4 V, respectively (Fig. 2c). The molten salt cathode delivered an average discharge specific energy of  $155 \text{ Wh kg}^{-1}$  (based on the weight of cathode materials) with an average energy efficiency of 83% (Fig. 2e). Both the charge energy density and the discharge energy density remained stable during cycling, which also indicated the high cyclic performance of the whole cell. The XRD results (Fig. 1c) indicate the high



**Fig. 3.** (a) Voltage profiles of the discharge/charge cycles at various current densities of 5, 10, 20, 30 and 40 mA cm<sup>-2</sup> with a constant specific capacity of 50 mAh g<sup>-1</sup>. (b) Voltage as a function of specific capacity at various current densities of 5, 10, 20, 30 and 40 mA cm<sup>-2</sup> with a constant specific capacity of 50 mAh g<sup>-1</sup>.



**Fig. 4.** (a) SEM image of the pristine Al foil. (b) SEM image of the Al foil after the 1st discharge/charge cycle at 10 mA cm<sup>-2</sup> with a specific capacity of 100 mAh g<sup>-1</sup>. (c) and (d) SEM images of the Al foil after the 100th cycle at 10 mA cm<sup>-2</sup> with a specific capacity of 100 mAh g<sup>-1</sup>. Scale bar, 2 μm (a,b,d) and 10 μm (c).

stability of LLZTO tube during cycling as there is no obvious phase change before and after cycling. To test the safety of the cell in case the LLZTO tube breaks, a puncturing experiment was conducted. The solid electrolyte tube was full of molten AlCl<sub>3</sub>-LiCl salt (2 g), and the stainless steel case was full of molten lithium (1 g). After heating to 210 °C, the LLZTO tube was punctured, and molten lithium and molten salt was partly contacted. After puncturing, the surface temperature of the stainless steel case increased by only about 50 °C.

The cell was also assembled in the discharge state and part of LiCl was replaced by NaCl to lower the cost. The pristine cathode was composed of LiCl, AlCl<sub>3</sub> and NaCl (mole ratio, 4: 1: 1) and little amount of Li was used to connect the LLZTO and the current collector. The cell was firstly charged at 5 mA cm<sup>-2</sup> with a specific capacity of 200 mAh g<sup>-1</sup> (Fig. S2a). After the first charge process, the cell delivered a stable cycling

performance (Fig. S2b), which indicates that the future cells can be produced without metallic lithium to be handled. Like ZEBRA batteries, all the required active metal can be inserted as salt, therefore the assembling process of the battery can be conducted just in a drying room. The cost of raw materials can be also significantly reduced when the cell is assembled in the discharge state, because metal Li is not indispensable. The impedance test result (Fig. S4) shows that the resistance of the full cell is small and leads to a promising future for the application of this battery.

To smooth the fluctuation of intermittent renewable energy, high rate capability is desired for grid energy storage. The rate performance of the AlCl<sub>3</sub>-LiCl || LLZTO || Li cell was shown in Fig. 3a. The cell was cycled at various current densities from 5 mA cm<sup>-2</sup> to 40 mA cm<sup>-2</sup> with a constant discharge/charge capacity of 50 mAh g<sup>-1</sup>. The voltage profiles at the

current density of  $10 \text{ mA cm}^{-2}$  showed no obvious changes before and after the cell was cycled at a high current density of  $40 \text{ mA cm}^{-2}$ , indicating good rate capability of the cell. The discharge plateaus were 1.65, 1.61, 1.54, 1.46 and 1.34 V at current densities of 5, 10, 20, 30 and  $40 \text{ mA cm}^{-2}$ , respectively (Fig. 3b). Even at a high current density of  $40 \text{ mA cm}^{-2}$ , the energy efficiency of cell was still higher than 65%. The power density of the  $\text{AlCl}_3\text{-LiCl} \parallel \text{LLZTO} \parallel \text{Li}$  cell reached  $52 \text{ mW cm}^{-2}$  at the current density of  $40 \text{ mA cm}^{-2}$ .

The service life of current collector is also an important factor for rechargeable batteries using in the grid energy storage. In the  $\text{AlCl}_3\text{-LiCl} \parallel \text{LLZTO} \parallel \text{Li}$  full cell, the Al foil not only functions as positive current collector, but also participates in the reaction. To investigate the stability of the Al current collector, scanning electron microscopy (SEM) was conducted to display the morphology changes of the Al current collector before and after the cycling test. As shown in Fig. 4a, the pristine Al foil was relative smooth with some little protuberances in the surface. After the first discharge/charge cycle, the surface became rough with some corrosion pits (Fig. 4b), which possibly originated from the defects in the surface of the pristine Al foil. After the 100 cycles, there were many particles with the sizes from several micrometers to  $30 \mu\text{m}$  on the surface of the Al current collector (Fig. 4c and d). Energy Dispersive Spectroscopy (EDS) characterization shows that the particles were composed of Al with little amount of oxygen (Fig. S3). The generation of the micron-sized Al particles were partially caused by the inhomogeneous deposition/dissolution of Al during cycles, which possibly led to the loss of the active Al and the corrosion of the current collector. However, the average thickness of the Al foil did not change after the 100 cycles. Even though the thickness of the Al foil used as current collector was only  $35 \mu\text{m}$ , there was no preferable corrosion point. Therefore, the inhomogeneous deposition of Al is not a serious problem for practical applications.

In summary, we demonstrated a newly designed high temperature battery with molten salt cathode-solid electrolyte-molten lithium anode, which can be operated at a relative low temperature of  $210^\circ\text{C}$  with good safety and low cost. The full cell achieved a stable performance for 100 cycles (800 h) with an average energy efficiency of 83% at a current density of  $10 \text{ mA cm}^{-2}$ . We propose that the cathode material is not limited to  $\text{AlCl}_3\text{-LiCl}$  in such battery design; and the battery can be further improved by optimize the molten salt cathode systems in the near future. We anticipate that our  $\text{AlCl}_3\text{-LiCl} \parallel \text{LLZTO} \parallel \text{Li}$  cells provide new opportunity for large-scale energy storage solutions.

## Notes

The authors declare no competing financial interest.

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## Appendix A. Supplementary data

Supplementary data to this article can be found online at <https://doi.org/10.1016/j.ensm.2019.07.027>.

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